

Sugar Alcohol based Materials for Seasonal Storage Applications

# FP7 project - SAM.SSA

Sugar Alcohol based Materials for Seasonal Storage Applications

Workshop and Onsite Demonstration– CiCenergigune

Miñano, Alava, Spain





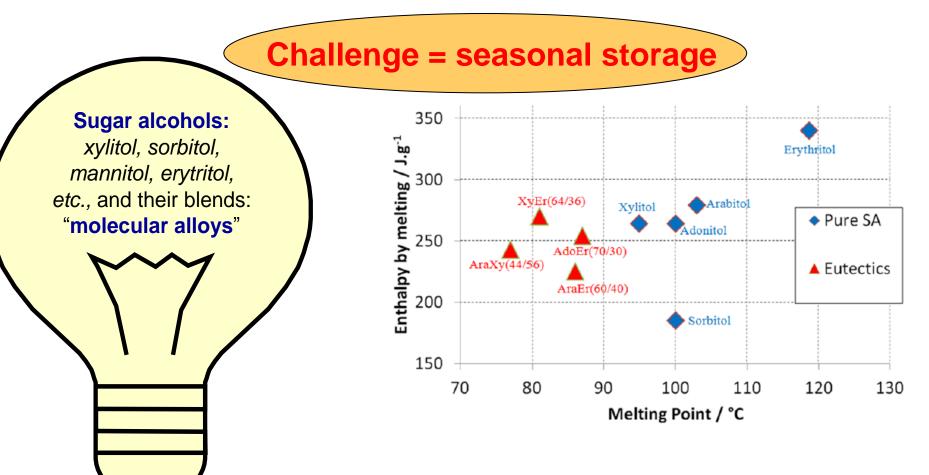
# Session 1, Sub Session: Heat Transfer Enhancement



Presented by: Alain CELZARD Contributions by: Prasanta JANA Vanessa FIERRO Alain CELZARD



Phase-change materials are investigated for thermal energy storage from the 70's, but ...





Sugar alcohols: xylitol, sorbitol,

*mannitol, erytritol, etc.,* and their blends: "**molecular alloys**" Context

Phase-change materials are investigated for thermal energy storage from the 70's, but ...

# Challenge = seasonal storage



High energy density: compact storage systems



High undercooling : long-term storage



Possible crystallisation by thermal shock or other means: heat recovery at the desired moment



Poor thermal conductors



# **Thermal conductivity improvement**

#### **DEVELOPMENT OF HIGHLY POROUS CARBON MONOLITHS**

• Open porosity higher than 80%

Thermal conductivity within the range 3 – 10 W/m/K

- Cost less than 6 €/kg
- Wide pore size for easy and in-depth impregnation with SA

Average thermal conductivities wanted !

A very unusual request ...
Why ?

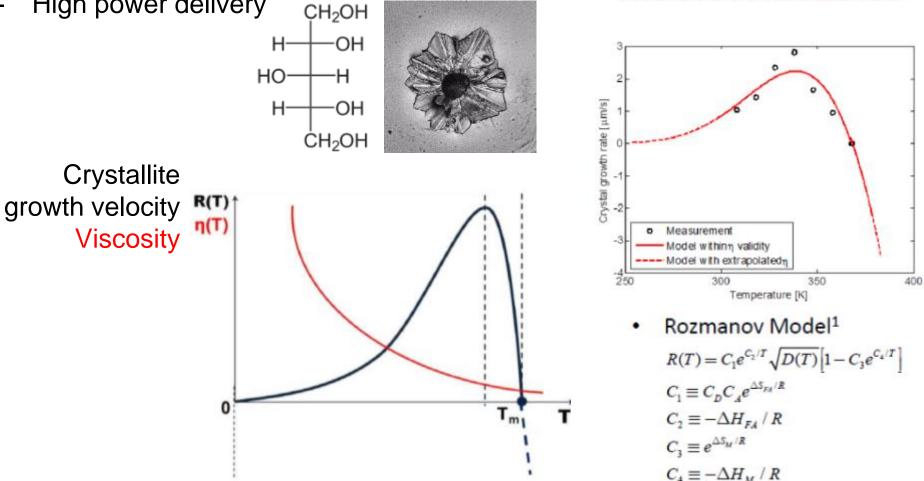


#### Context

XYLITOL'S PROPERTIES

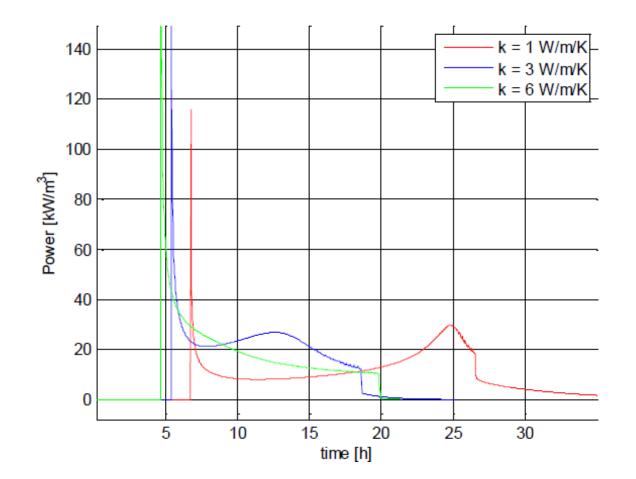
#### Need of a medium conductivity related to the optimisation between:

- Short time for heat delivery
- High power delivery





Effect of the thermal conductivity of the porous host on delivered power:





#### Porous carbon matrices have many advantages :



Cheaper than metal foams



Are much lighter than metal foams



Can have "green" precursors



Can be prepared with any porosity



Can have a broad range of physical properties



Can be functionalised and/or nanostructured





Require high-temperature treatment under inert atmosphere



## What kind of porous matrices ?

#### Most common carbon foams

**Reticulated vitreous carbon (RVC)** foams

(Derived from thermoset polymers)

Extremely low thermal conductivity Made from non-renewable resources 🙁 Expensive

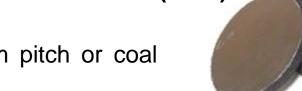
Duocel® RVC foam (ERG Corporation)

#### graphitic carbon (CGC) Cellular foams

(Derived from pitch or coal extracts)

Very high thermal conductivity

Made from non-renewable resources Prohibitive cost





POCOfoam® graphite foam (ORNL)



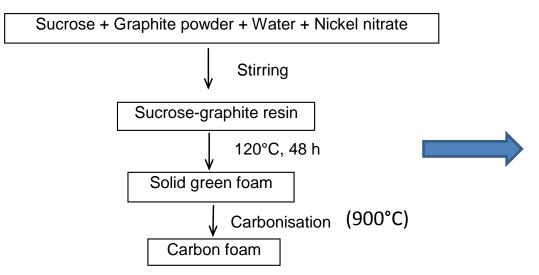
## What kind of porous matrices ?

#### CVC and RVC foams 10 Made from cheap (possibly renewable) resources $\begin{pmatrix} \circ \circ \\ & \end{pmatrix}$ Thermal conductivity (W/m/K $\begin{pmatrix} \circ & \circ \\ & & \end{pmatrix}$ Low cost C Low thermal conductivity Target area Our measurements for pure carbon foams ٥ 0.1 0.4 0.3 02 ۵5 Bulk density (g/cm<sup>3</sup>) Strategy :

### Preparing composite carbon foams containing a filler with a high thermal conductivity

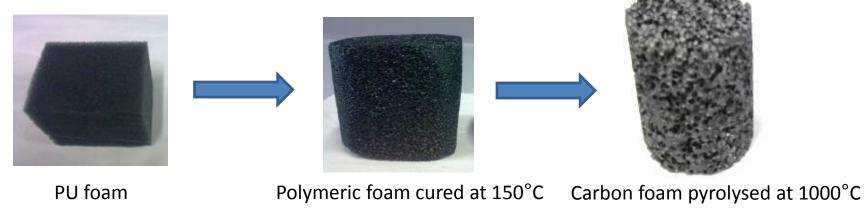


#### Autogenous foaming of sucrose solutions





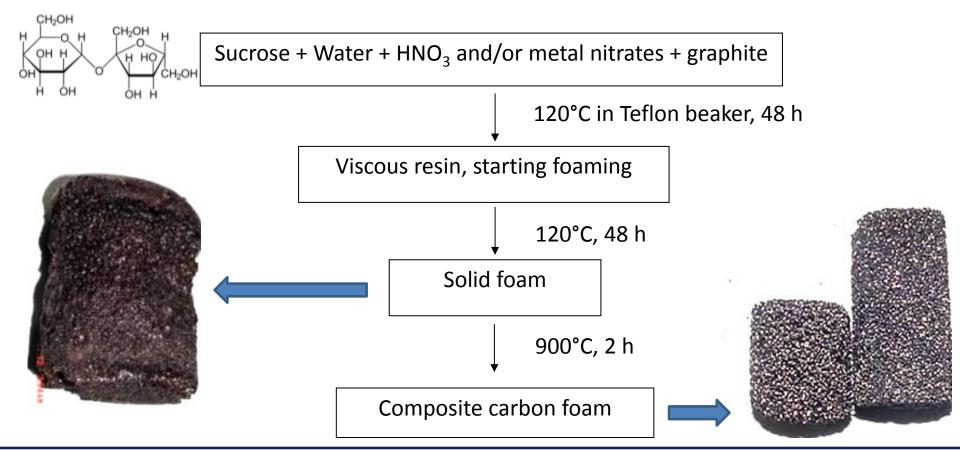
#### **Template method based on PU foams**





## Thermally induced decomposition and resultant foaming

Decomposition of a nitrate whereas a sucrose solution is polymerized in acidic conditions, leading to a foamed carbon precursor obtained at 120°C.





## Experimental design

#### **Reasons:**

- Need to maximise both porosity and thermal conductivity, whose changes are antogonistic
- Save time, cost and materials

#### **Principles :**

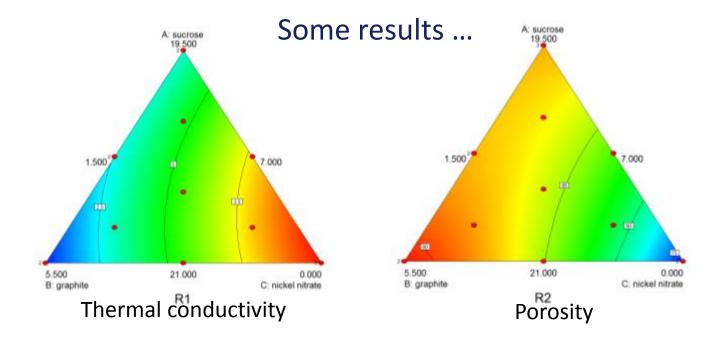
- Choosing the most relevant parameters to be changed for preparing as few samples as possible
- From the experimental results and the corresponding modelling, being able to predict the properties of any other formulation

#### Implementation:

- Choice of a graphite and a catalyst leading to the best results: a natural graphite having the lowest available particle size on one hand, and nickel nitrate on the other hand
- Use of Design-Expert 8.0.7.1. software for optimising the amounts of all ingredients, using porosity and thermal conductivity as responses.



#### Context



From the model:

R1= - 0.6536A - 4.61697B - 9.45242C + 0.34024AB + 0.56745AC + 0.16269 BC

R2= 8.13738A + 45.31808B - 125.15084C - 2.81767AB + 4.26041AC + 4.86517 BC

If A = 20, B = 6, C = 1: R1 = 2.9277 (model) versus 3.1 W/m/K (experiment) R2 = 85.784 (model) versus 85.8% (experiment)



## Same procedure applied to much bigger samples :

Slightly different ...

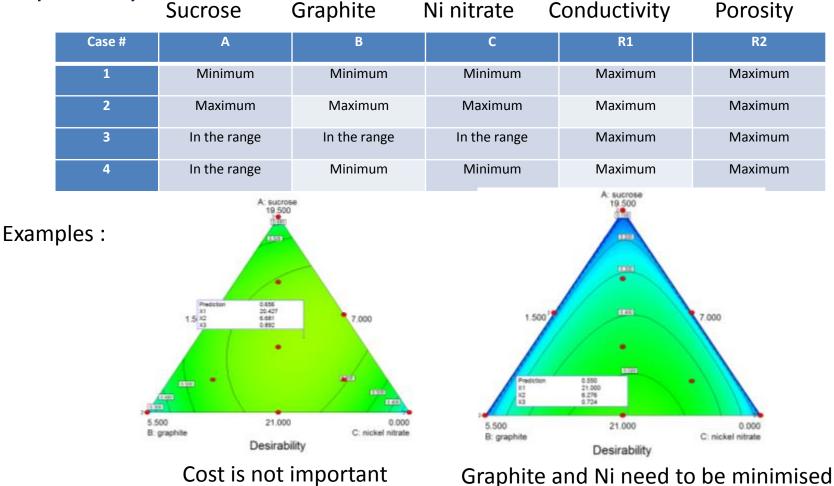
R1= 0.016522 A + 0.049148 B + 1.17358 C + 0.00151433 AB - 0.012735 AC - 0.022849 BC R2= 0.92073 A + 1.40489 B - 3.7681C - 0.016203 AB + 0.052576 AC + 0.08671 BC

... but more favourable results : less graphite and less nickel are required for getting the same results !

If A = 80, B = 20, C = 2: R1 = 2.9144 (model) versus 3.1 W/m/K (experiment) R2 = 85.664 (model) versus 85.8% (experiment)

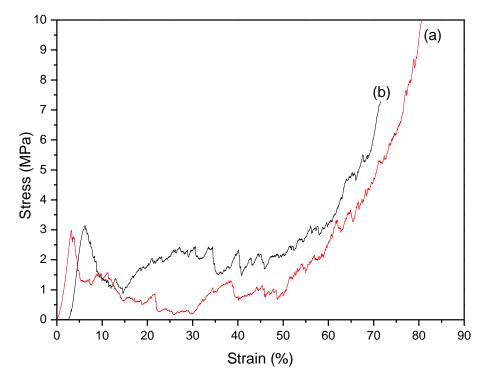


# One step further : more constraints for maximising both conductivity and porosity





## Mechanical properties of optimised foams :



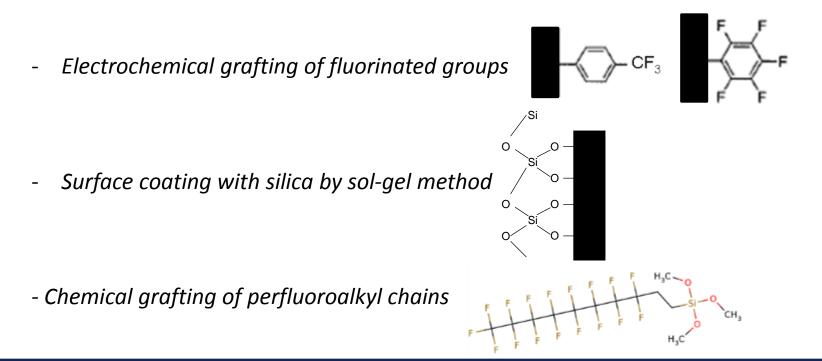
Composite foam	Compressive strength (MPa)	Elastic modulus (MPa)
Small sample	3.0	107.5
Big sample	3.1	107.9



## Hydrophobisation of optimised foams :

#### Different techniques used :

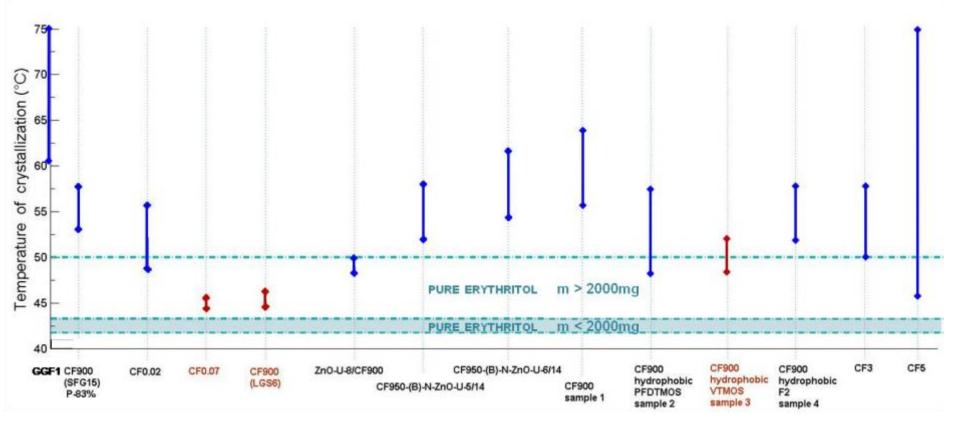
- Direct chemical fluorination  $\rightarrow$  Foams of overall composition  $CF_{0.04} - CF_{0.08}$ 





## Effect of hydrophobisation on subcooling :

Some foams promote nucleation (no hydrophobic effect), others hinder nucleation (hydrophobic)





## **Conclusion :**

Composite, graphitised, carbon foams made from sucrose and natural graphite are :

- cheap (~ 3 €/kg)
- highly porous (86%)
- easy to impregnate with sugar alcohols



- moderately conducting before ( $\lambda \sim 3$  W/m/K) and after ( $\lambda \sim 3.5$  W/m/K) impregnation with sugar alcohols
- resistant (3 MPa) and stiff (100 MPa) before impregnation
- even more after impregnation: 10 et 200 MPa, respectively
- promote or delay heterogeneous nucleation, depending on surface functionalisation



# WP 4 (MASA Thermal Conductivity Improvement)



Presented by: Dr. Mani Karthik Work Package Leader: Dr. Alain Celzard Participants: Dr. Bruno D'Aguanno and Dr. Abdessamad Faik

#### **Graphite Composite Foam by Template Method** SAM SSA

### Tasks:

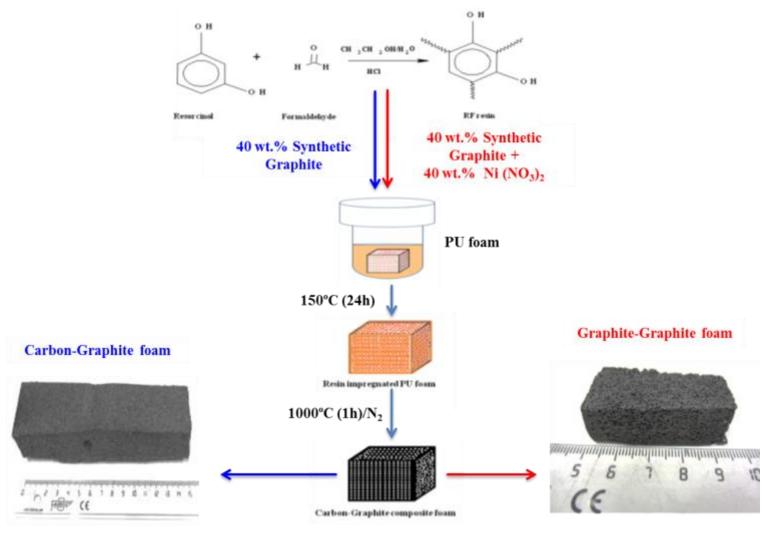
- Synthesis of graphite composite foam
- Structural characterization
- Impregnation of the matrix with PCMs (Erythritol)
- Thermophysical characterization
- **Chemical stability of the PCMs with carbon matrix**  $\checkmark$ Target
- Porosity: 85 90 %
- Density: 0.25 0.35 g/cm<sup>3</sup>
- Thermal conductivity: 2 4 W.m<sup>-1</sup>.K<sup>-1</sup>
- Cost: 6 7 €/Kg
- Mechanical stability
- Hydrophobic  $\succ$

## **Results Achieved**

- ✓ Porosity: 70 88 %
- Density: 0.20 0.45 g/cm<sup>3</sup>
- ✓ Thermal conductivity: 2 4 W.m<sup>-1</sup>.K<sup>-1</sup>
- ✓ Cost: 3 7 €/Kq
- ✓ Mechanical stability : 0.4 1.3 MPa
- Hydrophobic: contact angle 120°  $\checkmark$

# samesa Preparation of Graphite Composite Foams

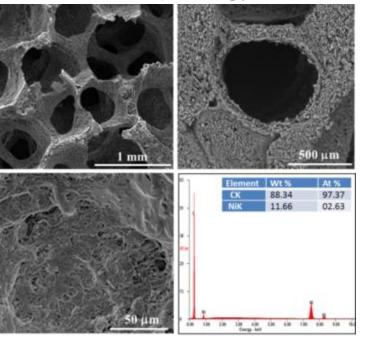
#### Preparation of Carbon-Graphite or Graphite-Graphite Composite Foams





# **Graphite Composite Foam**

#### Surface Morphology - SEM



#### Properties of the Carbon Foams

Samples	Composition Polymer/Graphite/Ni (wt. %)	Bulk Density (g/cm³)	Total Porosity (%)
Carbon foam (CF1)	100/0/0	0.1500	93.3
Carbon-Graphite foam (CGF1)	60/40/0	0.2338	88.86
Carbon-Graphite foam (CGF2)	50/50/0	0.3768	82.06
Graphite-Graphite foam (GGF1)	60/40/40	0.4252	79.75

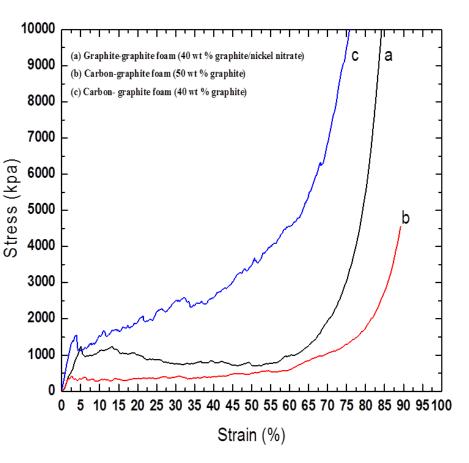


#### Target: Porosity: 85-90 %



# Compressive stress-strain behavior of the carbon foams

(GGF1)



Samples	Composition Polymer/Graphite/Ni (wt. %)	compressive strength (MPa)	Elastic modulus (MPa)
Carbon-Graphite foam (CGF1)	60/40/0	1.30	54
Carbon-Graphite foam (CGF2)	50/50/0	0.40	24
Graphite- Graphite foam	60/40/40	0.42	23

#### **Reticulated Vitreous Carbon (RVC)**

Commercial "Ultramet" carbon foam (0.16-0.76 MPa) Commercial "ERG" carbon foam (0.10-0.52 MPa)

#### Compression Strength

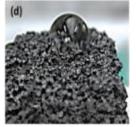
(0.10-0.76 MPa)

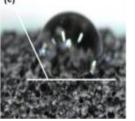
*Typical compressive stress-strain behavior of the composite carbon foams* 



### Surface Functionalization of Carbon Foams -Hydrophobic Carbon Foam

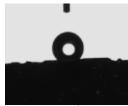
#### Water droplet on the surface of the carbon foam





#### **Graphite Foam** Surface coating with polytetrafluoroethylene (PTFE)

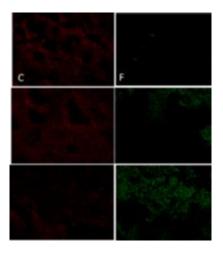
#### **Contact Angle Measurement**

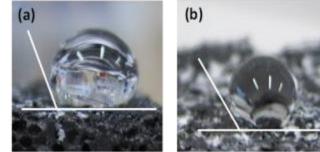


#### Contact Angle = 120°

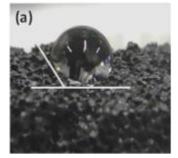
(c)

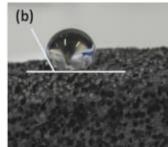
#### **EDAX Mapping**

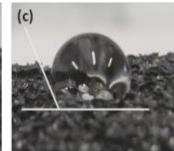


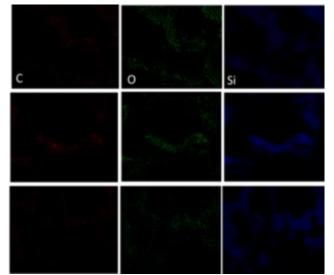


#### Surface coating with tetraethyl orthosilicate (TEOS)









# samesa Preparation of Erythritol-Graphite Composite Foam

## **Sugar Alcohol Infiltration**

- The erythritol infiltration was performed by placing a piece of carbon foam at the bottom of a pool of liquefied erythritol in a beaker/glass plate.
- > After infiltration, the foam piece was removed from the beaker and allowed to cool down.
- The excess of erythritol in the outer surface of the carbon foam was removed by carefully scraping with a blade or knife.

#### Impregnation of Sugar Alcohol



Sugar alcohol

Molten Sugar alcohol

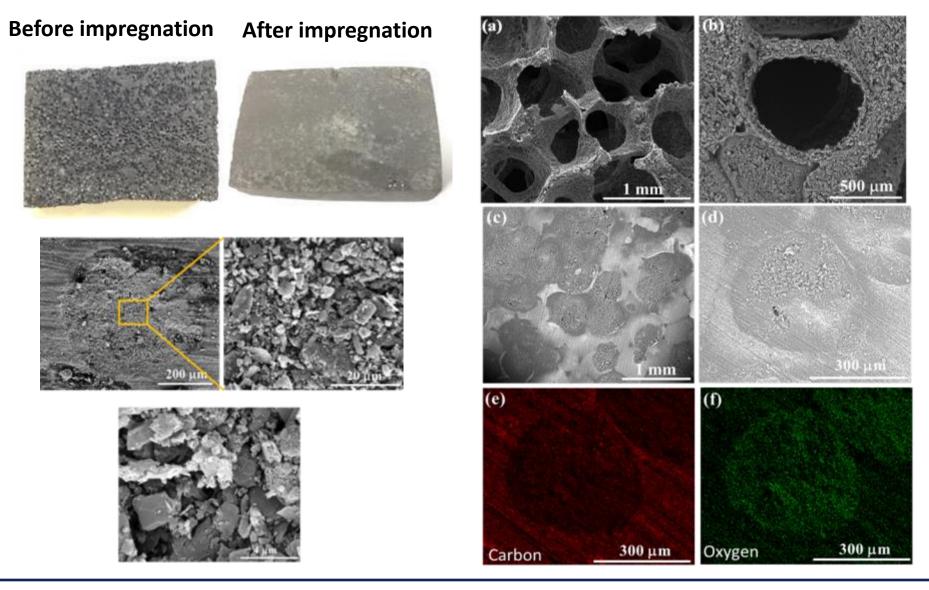


Molten Sugar alcohol + Foam

Infiltration of Sugar alcohol is ca. 70-80 wt % depends on the porosity of the carbon foam



# Structure of Graphite Foam Before and After Erythritol Impregnation

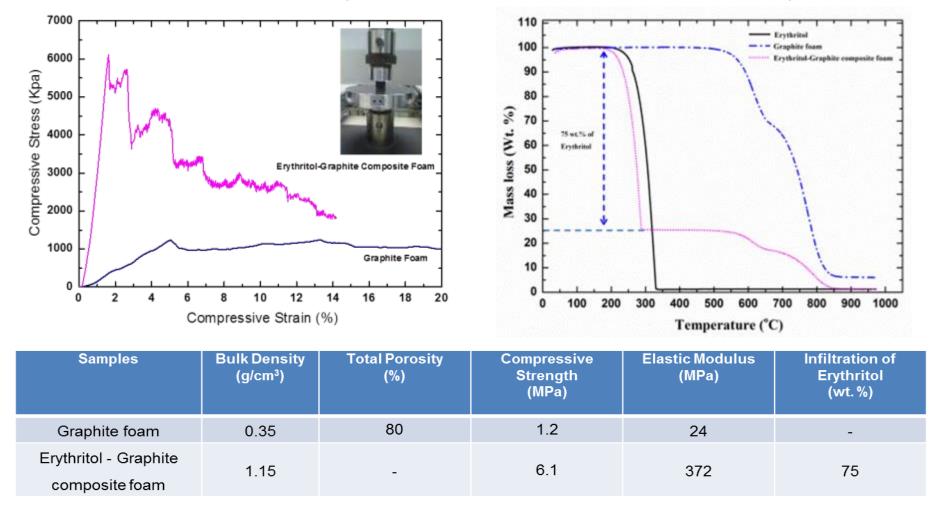




# **Characteristics of the Materials**

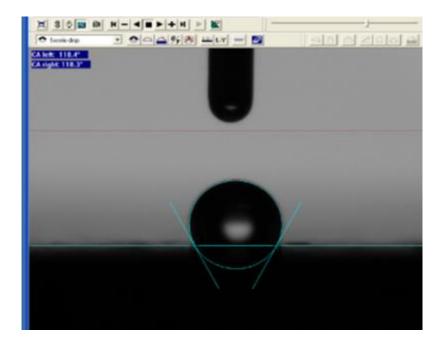
**Mechanical Stability** 

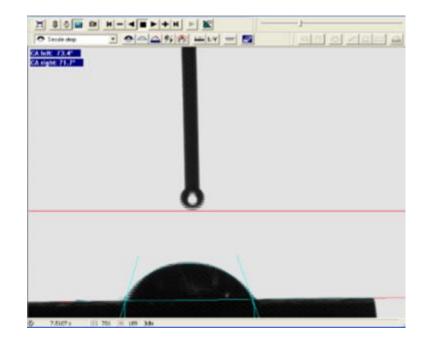
**Thermal Stability** 





# **Contact Angle Measurement**





Graphite composite foam Hydrophobic

#### Erythritol-Graphite Composite Foam Hydrophilic

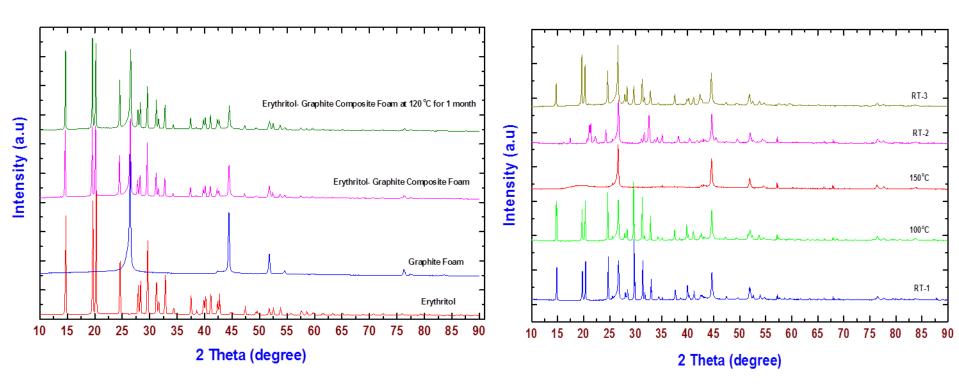


## Chemical Stability of the PCM within the Carbon Matrix

# **Compatibility between Erythritol and Graphite Foam**

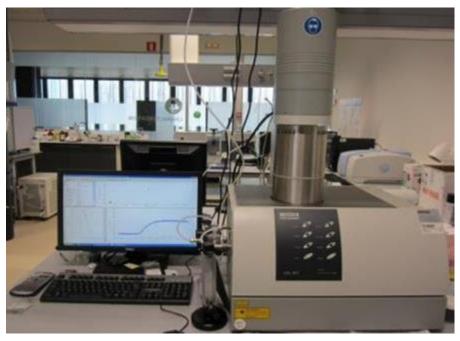
**XRD Patterns** 

In-situ XRD Patterns (In-situ heating chamber)





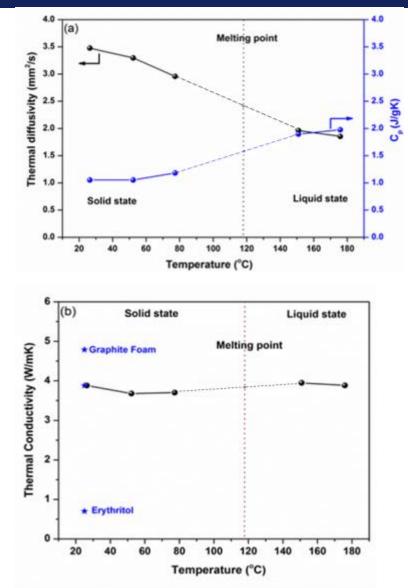
# Thermal Conductivity Enhancement of PCM Composite



Laser Flash Apparatus (LFA 457, Netzsch)

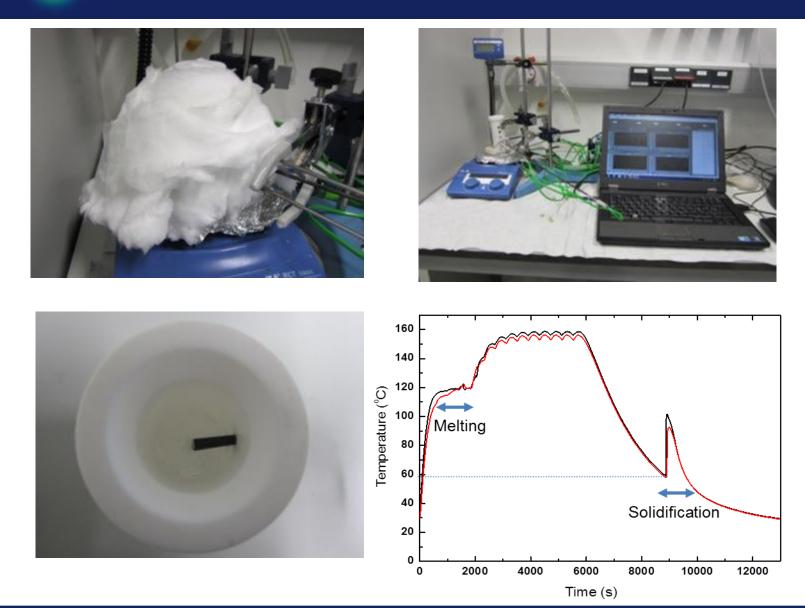
 $\mathbf{k} = \alpha \cdot \rho \cdot C_p$ 

where k is the thermal conductivity in W/mK  $\alpha$  is the thermal diffusivity in mm<sup>2</sup>/s  $\rho$  is the density in g/cm<sup>3</sup> Cp is the specific heat in J/g.K





# **Small Experimental Device**





# Conclusions

- Carbon-graphite and graphite-graphite composite carbon foams were prepared by template method using commercially available polymeric foams (PU) as macroporous sacrificial template.
- Erythritol-graphite foam as a stable composite PCM was obtained by a simple incipient wetness impregnation method with high impregnation ratio.
- ✓ The in-situ XRD result clearly indicates that there is no chemical reaction between erythritol and graphite and thus the composite has not only good structural stability but also good compatibility between erythritol and graphite composite foam.
- ✓ It was found that the thermal conductivity of the erythritol-graphite composite foam (3.88 W/mK) was enhanced ca. 5 times higher as compared with that of pristine erythritol (i.e. 0.7 W/mK). This enhancement can significantly reduce the charging and discharging times of the PCM storage system.
- ✓ The PCM/foam composite has a melting point of 118 °C and latent heat of ca. 250 J/g which corresponds to the mass percentage (ca. 75 wt. %) of the erythritol within the composite foam.
- It can be concluded that the carbon matrix is not affecting the melting behaviour and enthalpy of the erythritol.